

Plane wave/pseudo potential calculations in solid state — II

Ari P Seitsonen

IMPMC, CNRS & Universités 6 et 7 Paris, IPGP

École thématique calcul/RMN

Different basis set

Practical calculations — What to control
Quantum Espresso/PWSCF

Localised basis sets

Comparison: Localised basis sets vs plane waves

Summary

- 1 Different basis set
 - Localised basis sets
 - Comparison: Localised basis sets vs plane waves
- 2 Practical calculations — What to control
- 3 Quantum Espresso/PWSCF

Expansion using a basis set

- For practical purposes it is necessary to expand the Kohn-Sham orbitals using a set of basis functions
- Basis set $\{\varphi_\alpha(\mathbf{r})\}_{\alpha=1}^M$
- Usually a linear expansion

$$\psi_i(\mathbf{r}) = \sum_{\alpha=1}^M c_{\alpha i} \varphi_\alpha(\mathbf{r})$$

Different basis sets

- Plane waves
 - quantum espresso/PWSCF, CPMD, abinit, paratec, VASP, CASTEP, ...
- Gaussian functions
 - TurboMole, MolPro, GaussianXX, ...
- Slater functions
 - ADF
- Numerical/ δ basis
 - DMol3, numol, ...
- Mixtures
 - CP2k, Wien2k, ...

Atomic orbital basis sets

Philosophy

Molecules are assemblies of slightly distorted atoms

$$\varphi_{\alpha}(\mathbf{r}) = \varphi_{\alpha}(r) Y_{lm}(\Theta, \phi)$$

$$\varphi_{\alpha}(r) = \begin{cases} \exp[-\alpha r^2] & \text{Gaussian} \\ \exp[-\alpha r] & \text{Slater} \end{cases}$$

$\varphi_{\alpha}(\mathbf{r}; \mathbf{R}_I)$: basis functions are attached to nuclear positions

Advantages / disadvantages of AO basis sets

- + according to chemical insight
- + small basis sets give already good results
- non-orthogonal
- depend on atomic position
- basis set superposition errors (BSSE)

Properties of plane waves

$$\varphi_{\mathbf{G}}(\mathbf{r}) = \frac{1}{\sqrt{\Omega}} \exp[i\mathbf{G} \cdot \mathbf{r}]$$

- Plane waves are **periodic** wrt. box **h**
- Plane waves are **orthonormal**

$$\langle \varphi_{\mathbf{G}'} | \varphi_{\mathbf{G}} \rangle = \delta_{\mathbf{G}', \mathbf{G}}$$

- Plane waves are **complete**

$$\psi(\mathbf{r}) = \psi(\mathbf{r} + \mathbf{L}) = \frac{1}{\sqrt{\Omega}} \sum_{\mathbf{G}} \psi(\mathbf{G}) \exp[i\mathbf{G} \cdot \mathbf{r}]$$

Comparison to AO basis set

Plane Waves:

$$\begin{aligned} \frac{1}{2} G^2 &< E_{\text{cut}} \\ \frac{1}{2} G'^2 &< E_{\text{cut}} \\ \frac{1}{2} (G + G')^2 &< \left(\sqrt{E_{\text{cut}}} + \sqrt{E_{\text{cut}}} \right)^2 = 4E_{\text{cut}} \end{aligned}$$

Atomic orbitals: Every product results in a new function

$$\varphi_{\alpha}(\mathbf{r} - \mathbf{A}) \varphi_{\beta}(\mathbf{r} - \mathbf{B}) = \varphi_{\gamma}(\mathbf{r} - \mathbf{C})$$

Linear dependence for plane waves vs. quadratic dependence for AO basis sets.

Summary

- 1 Different basis set
- 2 **Practical calculations — What to control**
 - Starting a new calculation
 - Useful tools
 - Examples of convergence tests
- 3 Quantum Espresso/PWSCF

How to start a new calculation

Initial considerations

- What is the best method/code available?
(classical/approximate/DFT potential)
- Size of system? (computational resources available)
- Static or dynamics calculation?

Procedure for DFT calculations

- Choice of the system (number of atoms, unit cell, ...; can be modified later)
- Test for static computational parameters
 - description of nuclei (CPMD: pseudo potentials)
 - basis set (CPMD: cut-off energy only)
 - computational unit cell for isolated systems (molecules)
 - exchange-correlation functional

Choice for computational parametres: Unit cell

- In plane wave calculations one *always* has to define a computational unit cell, even for isolated objects such as atoms or molecules
 - It should be large enough so that the electronic wave functions and density vanish at the boundary of the box, or that the periodic images of the molecule do not interact with each other
 - Test simply by increasing the size of the box systematically until the main parametres of interest (energies, vibrational frequencies, Kohn-Sham eigenvalues, ...) do not change anymore
- In naturally periodic systems like solids, liquids, slab geometries (surfaces, 2-dimensional periodicity), polymers (1-dimensional periodicity) one has to test for the convergence with respect to k point sampling

Choice for computational parametres: XC functional

- Does the system size allow for the use of hybrid functionals?
- If not, would LDA provide better accuracy than GGA's? This is normally the case only in solids or other densely packed materials, for the bulk properties. Also used sometimes in cases where GGA's fail qualitatively (eg adsorption of molecules with phenyl ring on transition metal surfaces)
- If GGA, which one of them? BLYP (and BP86) is favoured by chemists, PBE by physicists
- Other functionals: Meta-GGA's, asymptotically corrected functionals, self-interaction corrected functionals, ...

Choice for static computational parametres: PP

Considerations when selecting a pseudo potential

- Which valence configuration? Semi-core (= state in between “clear” core and valence states; normally $-20 \dots -50$ eV below the vacuum level) in core or valence?
- Do the core and valence electron density over-lap considerably? If yes, is NLCC necessary?
- Type of pseudo potential? Troullier-Martins, Vanderbilt, other norm-conserving one, ...
- Kleinman-Bylander scheme or Gauß-Hermite integration?
- Which l_{\max} , l_{loc} ? The former is more an issue of computing time, the second crucial to avoid
- Could the core radii be larger (to reduce basis set/computing time)?

Choice for computational parametres: PP testing

- Select a (small) representative system, preferably several of them; for example of different bonding or hybridisation types (eg purely covalent C-C, partially ionic/polarisation C-N or C-H, ionic like H-F)
- Start with a large basis set (please see next topic) in order to be sure of convergence
- Compare to all-electron or other, reliable pseudo potential calculation employing the same XC functional, if possible; comparisons with experiments have a second priority, due to the known, systematic errors in the XC functionals
- Quantities to compare: Binding energies, bond lengths, angles, vibrational frequencies, ...
- The differences to proper all-electron calculations should be of the order of 0.01 \AA or less in bond lengths, $< 50 \text{ meV}$ for binding energies, vibrational frequencies $< 20 \text{ cm}^{-1}$

Choice for computational parametres: PP parametres

If you need to optimise the parametres of the pseudo potential like the core radii, NLCC radius *etc*, you have to change them and re-run these tests; again until the property of interest does not change anymore significantly (Please remember: It is always nicer to be on the safe side. . .)

Choice for computational parametres: Basis set

- With plane wave basis set only one parametre: Cut-off energy E_{cut} ; units Rydberg atomic units (historical / convenience reasons: $E_{\text{cut}} [\text{Ha}] = \frac{1}{2} |\mathbf{G}_{\text{max}}|^2 = \frac{1}{2} E_{\text{cut}} [\text{Ry}] \Rightarrow |\mathbf{G}_{\text{max}}| [1/\text{Bohr}] = \sqrt{E_{\text{cut}} [\text{Ry}]}$)
- You can start from a low value of E_{cut} for fast convergence, increase it in steps of 5, 10 or 20 Ry until the properties of interest converge; please notice that the total energy of the system
- Typical values
 - 2p elements, transition metals, Troullier-Martins pp's: 50-90 Ry
 - 3p, 4p elements, alkali metals with NLCC, transition metals of the 5p row *etc*, Troullier-Martins pp's: 20-40 Ry
 - Vanderbilt pp's: 20-35 Ry

Useful tools

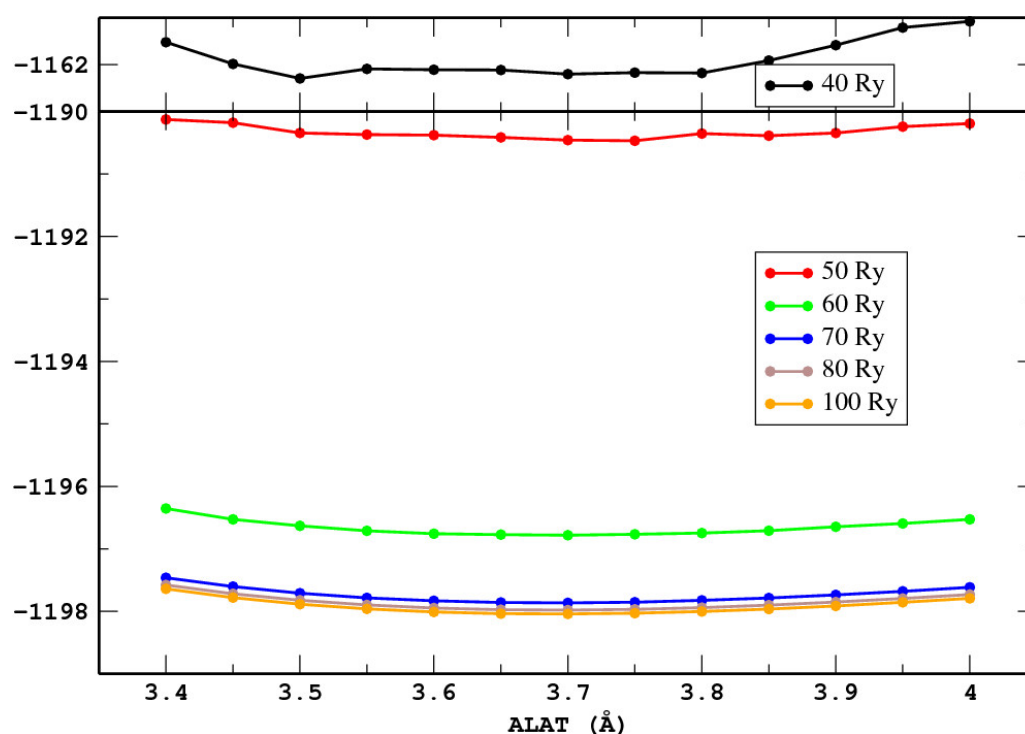
- `units` command: Conversion between different units:

```
localhost (~) : units
2084 units, 71 prefixes, 32 nonlinear units

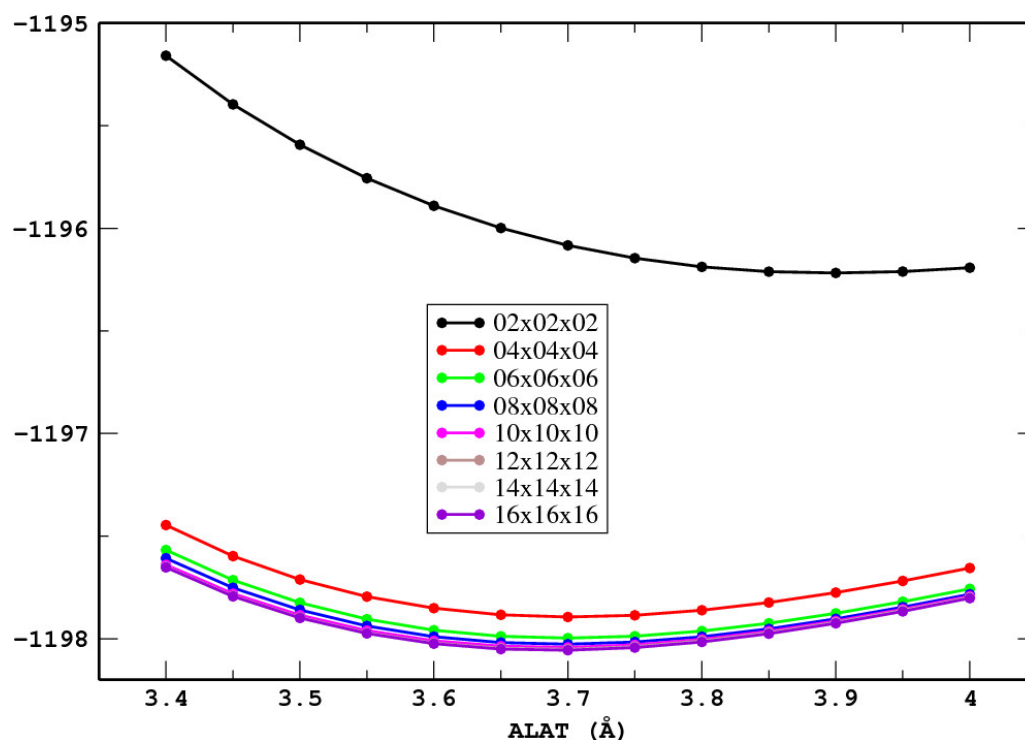
You have: 32*(15.9994+2*1.0067) atomicmassunit / (9.885 angstrom)^3
You want: g cm^-3
          * 0.99094647
          / 1.0091362
You have:
```

- `openbabel` converts between different graphics file formats
- `xmgrace` or `gnuplot` for 2-dimensional graphics
- `xcrysden` understand input and output of PWSCF
- graphics programs for 3-dimensional plots and atomic configurations, movies: `gOpenMol vmd`, `xmalemol`, ...

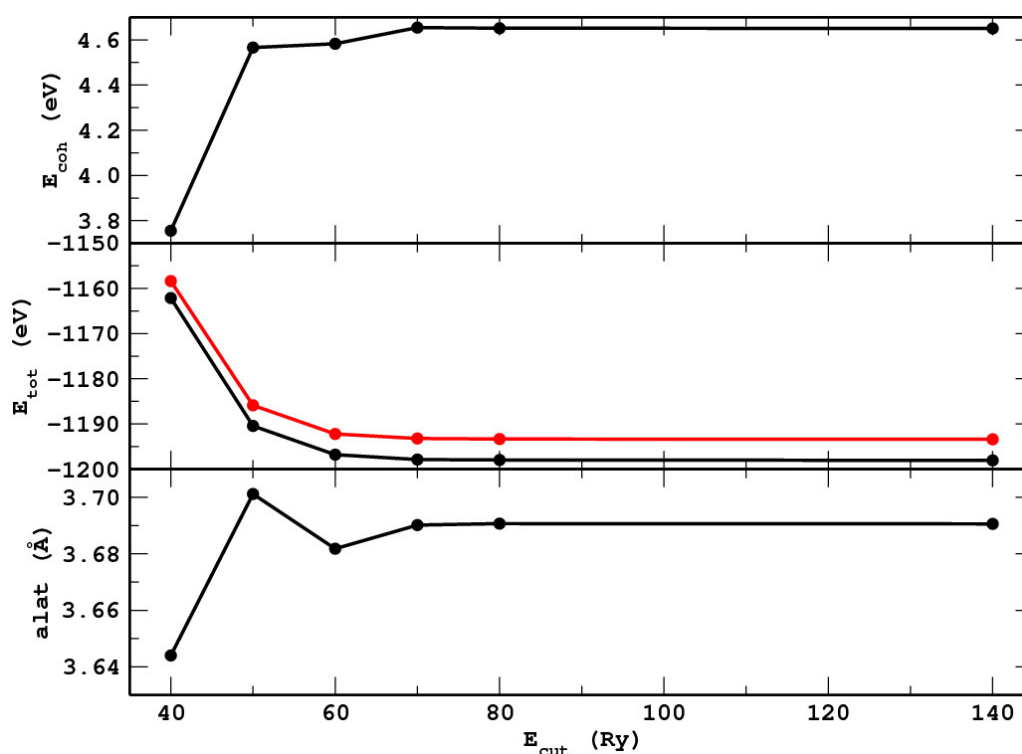
Convergence tests: Cut-off energy



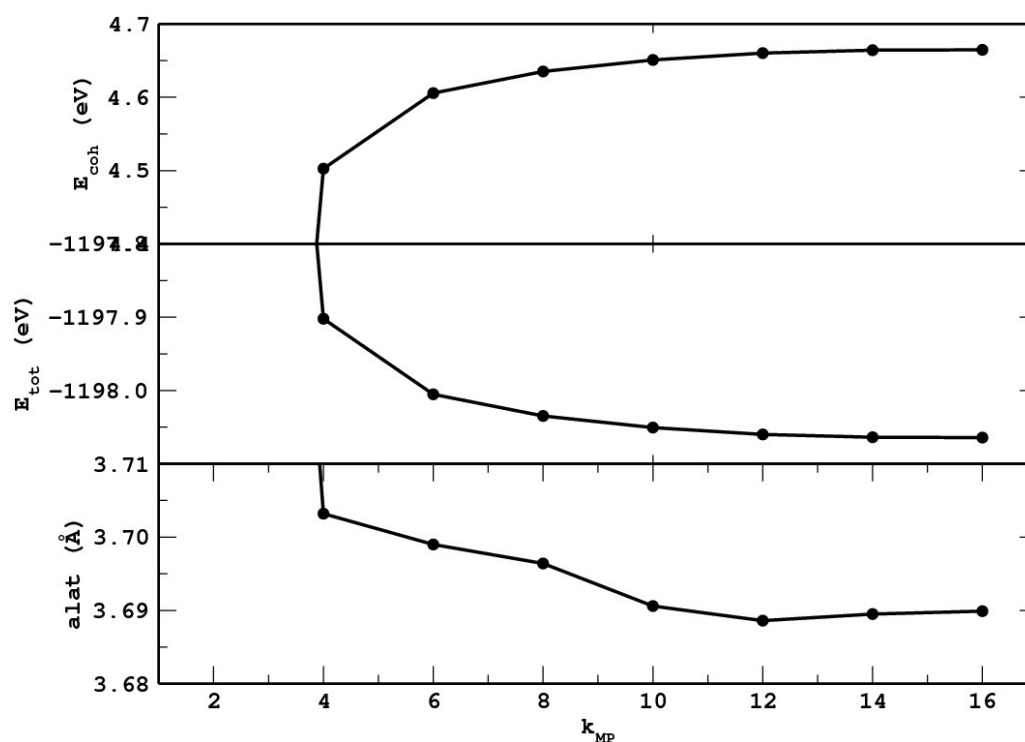
Convergence tests: k point set



Convergence tests: Cut-off energy



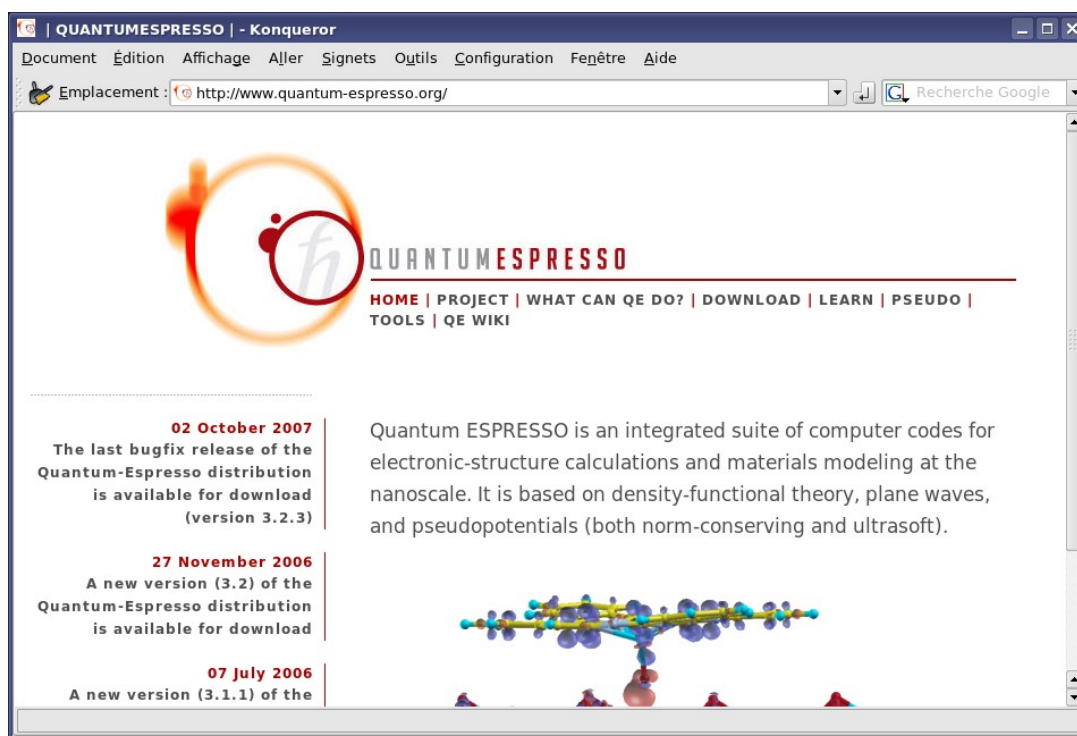
Convergence tests: k point set



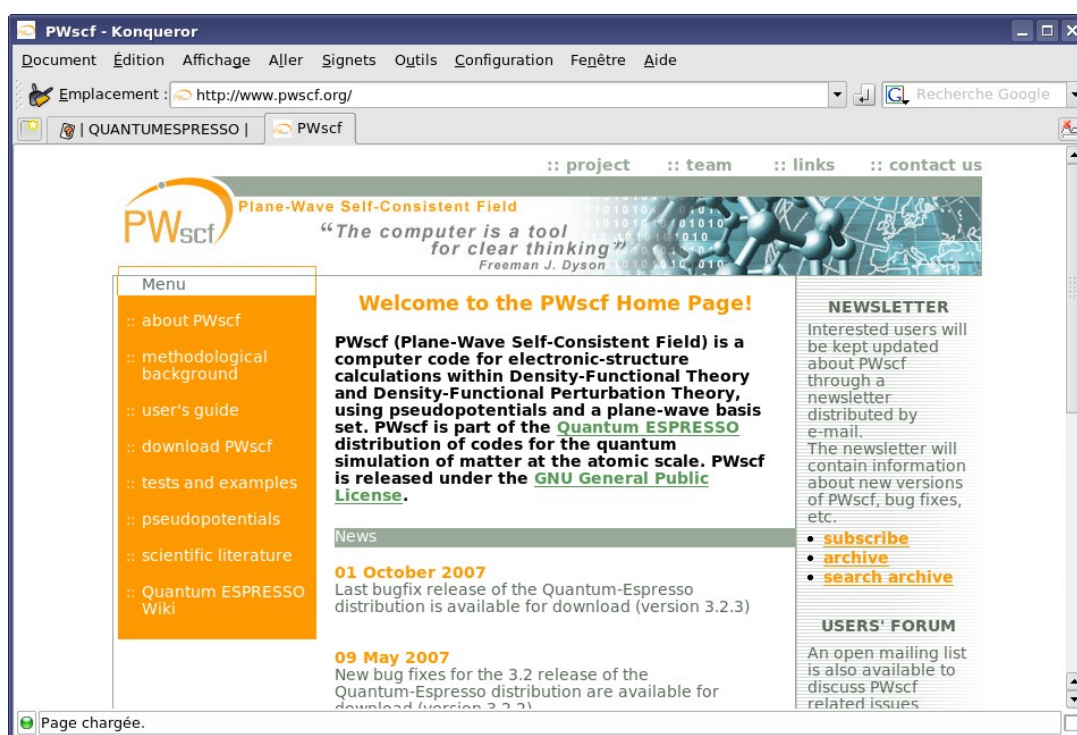
Summary

- 1 Different basis set
- 2 Practical calculations — What to control
- 3 **Quantum Espresso/PWSCF**
 - Introduction
 - Basic usage
 - Example

Quantum Espresso <http://www.quantum-espresso.org/>



PWSCF <http://www.pwscf.org/>



Quantum Espresso

- Historically started from “original” Car-Parrinello code
- Joined recently from PWSCF, CPV and FPMD

Running the executable

- Machine-dependent!
- Serial run:

```
./pw.x -input my-input-file > my-output-file &
```
- Parallel run, linux

```
mpirun -np ${NSLOTS} ./pw.x -npool 4 -input my-input-file > my-output-file &
```
- Parallel run, LoadLeveler:

```
./pw.x -npool 4 -input my-input-file > my-output-file &
```

Example: fcc-Cu

```
&control
  calculation='scf'
  restart_mode='from_scratch',
  pseudo_dir = '/home/seitsonen/usr/espresso/PP_LIBRARY/',
  wfcdir='/tmp/'
  prefix='fcc-Cu'
  tstress = .true.
  tprnfor = .true.
/
&system
  ibrav = 2, a = 4.00, b = 4.00, c = 4.00,
  nat= 1, ntyp= 1,
  nbnd = 9
  ecutwfc = 140
!   ecutrho = 0
  occupations='smearing', smearing='fermi-dirac', degauss=.00367490107593722133
/
&electrons
  diagonalization='david'
  conv_thr = 1.0e-9
  mixing_beta = 0.8
/
ATOMIC_SPECIES
Cu 63.5500 Cu_pbe-20071125.UPF
ATOMIC_POSITIONS angstrom
Cu 0.0 0.0 0.0
K_POINTS (automatic)
16 16 16 0 0 0
```

Help on input

- Input is explained in file `Doc/INPUT_*`
- One can ask for help on the Mailing list
- Different values for 'ibrav/celldm' are explained at the end of `Doc/INPUT_PW`
- Please be aware on units for lattice vectors and coordinates: Bohr, Ångström, a_0 , lattice vectors, ...